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The average crystal structure of  $\gamma$ -Na<sub>2</sub>CO<sub>3</sub>. By G.C.Dubbeldam and P.M.DE WOLFF, Delft University of Technology, Werkgroep Microstructuur, Lorentzweg 1. Delft, Netherlands

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 $\gamma$ -Na<sub>2</sub>CO<sub>3</sub> is a deformed version of the hexagonal  $\alpha$ -phase structure. This was shown by successful refinement of a corresponding trial structure (symmetry C2/m) to R=12.8% for 313 independent reflexions, the indices *hkl* being integers. The satellite reflexions which correspond to the actual modulated structure were not included in this refinement, so that it refers to the average structure only. The main feature of the modulation is apparent from the fact that the average structure contains two  $CO_3$  ions with half atoms. The modulation can, accordingly, be described as an ordered distribution of anions over two orientations.

Though it is one of the most common compounds, Na<sub>2</sub>CO<sub>3</sub> is still a white patch on the map of crystal structures. The present analysis shows a possible cause for this remarkable fact: the room temperature or  $\gamma$ -phase possesses a modulation anomaly, so unexpected in such a chemically trivial compound that it may have baffled early workers and just been forgotten afterwards.

At room temperature, Na<sub>2</sub>CO<sub>3</sub> crystallizes in a Ccentred monoclinic lattice with the following parameters (23°C):

$$a = 8.90, b = 5.24, c = 6.04 \text{ Å}, \beta = 101.2^{\circ}; Z = 4$$
.

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This phase was termed the y-phase by Brouns, Visser & de Wolff (1964; 'BVW'). It shows up to 6 sharp satellites for each normal reflexion, corresponding to rows in the reciprocal lattice. The row containing the normal reflexion *hkl* is given by

i) 
$$h = ha^* + kb^* + lc^* + mp^*$$
 (1)

where *m* is a small integer and  $\mathbf{p}^* = 0.182\mathbf{a}^* + 0.318\mathbf{c}^*$ .

Further properties of these satellites are:

(ii) With increasing temperature, the fractional components of p decrease continuously from the above values to, e.g.  $\mathbf{p}^* = 0.154\mathbf{a}^* + 0.286\mathbf{c}^*$  at 300 °C.

### Table 1. Atomic coordinates for y-Na<sub>2</sub>CO<sub>3</sub>

First three columns: final averaged structure. Last three columns:  $\alpha$ -Na<sub>2</sub>CO<sub>3</sub> on orthohexagonal axes.

		x	У	Z	x	У	z
$4 \times \frac{1}{2}$ Na(1)	g	0.000	0.018	0.000	0.000	0.000	0.000
$4 \times \frac{1}{2}$ Na(2)	h	0.000	0.022	0.200	0.000	0.000	0.200
$8 \times \frac{1}{2}$ Na(3)	j	0.172	0.544	0.749	0.167	0.200	0.750
$8 \times \frac{1}{2}C$	j	0.163	0.488	0.251	0.167	0.200	0.250
$8 \times \frac{1}{2}O(1)$	j	0.123	0.255	0.314	0.101	0.300	0.250
$8 \times \frac{1}{2}O(2)$	j	0.291	0.491	0.173	0.300	0.200	0.250
$8 \times \frac{1}{2}O(3)$	j	0.079	0.676	0.255	0.101	0.700	0.250

# Table 2. Measured and calculated structure factors

# The columns are L, $F_o$ , $F_c$ .

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(iii) The (h0l) plane k=0 is free from satellites.

These findings, reported without comment in BVW, can easily be translated into real space:

Re (i). Such a diffraction image is the well-known result of a modulation of the structure. A modulation can occur as a substitutional type, such as described recently by Jamieson, de Fontaine & Abrahams (1969). In Na<sub>2</sub>CO<sub>3</sub>, however, such a defect seems hardly possible. The only remaining possibility is that of a displacement modulation. This means that the coordinates of the *j*th atom in the unit cell q, r, s, (taken from a fixed origin) can be written:

$$x_{jqrs} = \bar{x}_j + x_j^d(w_{qrs})$$

with similar equations for y and z. Here  $\bar{x}_i$  is a fixed average coordinate value, whereas the displacement component  $x_i^{d}$  is a periodic function, averaging zero and with period 1, of

$$w_{qrs} = \mathbf{p^*} \cdot (q\mathbf{a} + r\mathbf{b} + s\mathbf{c})$$
.

Thus planes of constant displacement  $\bar{x}_j^a$  are perpendicular to **p**<sup>\*</sup>, and repeat at a distance  $1/|p^*|$  which in the present case is 16.3 Å.

Re (ii). Description of the structure as normal (unmodulated) based on some super-cell cannot be exact, since such a description is in conflict with the continuous change of  $\mathbf{p}^*$  with temperature.

Re (iii). The displacements are parallel to **b**, since the *b*-projection is unmodulated. This means that the displacements must be parallel to the *p*\*-wave front. In terms of the modulation function it means  $x^d = z^d = 0$ .

The present analysis is based upon normal reflexions [m=0 in equation (1)].

The resulting electron density – provided phases are assigned correctly – is the average of the actual distribution over all unit cells. Hence we expect each atom to show up in this distribution as a peak elongated in the y direction, or even split up into two peaks, depending on the amplitude and wave form of its modulation function  $y_{f}^{d}$ . Accordingly, we started by looking mainly at the *b*-projection, which can be expected to be unambiguous.

As a first trial structure we used a model derived from the hexagonal  $\alpha$ -form (BVW; structure derived by Bijk & Wisser, 1969) because the  $\gamma$ -form, after loosing its satellites at 360°C, transforms into the  $\alpha$ -form at 489°C without discontinuities in the cell parameters. The coordinates of the  $\alpha$ -form with respect to orthohexagonal axes are given in Table 1. The space group C2 was arbitrarily chosen. Though eventually disproved (see below) it gave the correct symmetry in the (h0l) projection; a later trial with Cm did not lead to a good refinement.

The main uncertainty was the orientation of the CO<sub>3</sub> ions. Using step-wise refinement (Bhuiya & Stanley, 1963) successively for the Eulerian angles of this group and for the other coordinates, a reasonable agreement between  $F_o$  and  $F_o$  (at this stage from integrated precession photographs) was obtained, first for the h0l reflexions, then also for the h1l reflexions. Higher k values were not used at this stage because the strength of the satellites in high k layers is such that to aim at a high accuracy of y coordinates is pointless until an explanation of the satellites has been found. Further refinements were based on integrated Weissenberg photographs (taken with Mo K $\alpha$  radiation) of a crystal rotating about the b axis. The largest dimension of the crystal was ~0.5 mm; no absorption correction was applied. Atomic form factors were taken from Moore (1963), and the leastsquares refinement program used a single isotropic temperature factor. This led to a disagreement index  $R=\Sigma |F_o-|F_c||/\Sigma F_o=17\%$  for 102 h0/reflexions and 84 h1/reflexions including those not observed. The corresponding structure is shown in Fig.1.

Though all distances are reasonable - including those within a single  $CO_3$  ion – the *h*1*l* reflexions showed some large discrepancies. Moreoever, in all these cases  $Im(F_c)$ was large and  $|F_c| > F_o$ . Since the imaginary part arises from the departure of the structure from C2/m symmetry (mainly caused by the two O atoms almost coinciding in the bprojection) and since the good agreement for hol reflexions shows that projection to be essentially correct, the solution is clear: there must be two alternative orientations [related by an (010) mirror plane] for each  $CO_3$  ion. This hypothesis is corroborated by the drop in the R index to 14.35% for the same reflexions as before, just by doubling the anion in the way indicated. The last refinements (of the y coordinates only) have accordingly been performed in the space group C2/m. The R index became 12.8% using 102 hol, 84 h1l, 72 h2l and 55 h3l reflexions. For these refinements a program of the Bhuiya-Stanley type has been used; the layer scale factors were obtained from a least-squares program. The results are shown in Tables 1 and 2.



Fig. 1. Trial structure with C2 symmetry. Above: projection along b; below: projection along c. Dots: C; large circles: O; small circles: Na. Na and O atoms at or near y=0 are open, those at or near  $y=\frac{1}{2}$  are hatched in the b projection. The average structure with symmetry C2/m is obtained from this Figure by superposing it with its mirror image formed by mirror planes m at the levels indicated in the lower Figure.



Fig.2. Schematic projection of the anions: (a) in the actual modulated structure; (b) in the average structure. The projection direction is perpendicular to  $\mathbf{p}^*$  and  $\mathbf{b}$ , and the C-O(2) direction is supposed to be parallel to it so that C and O(2) coincide in the centre of each projected anion. As seen from Fig.1, this is not quite true.

The presence of two 'half' carbonate ions at once makes clear the main feature of the modulation: the satellites of the  $\gamma$ -phase allow us to explain the double atoms of the anion as the split atoms representing an atom in the average electron density distribution of the structure, as discussed above. If there had been no satellites, the double anion would have had to have been ascribed to disorder in the anion orientation (this is most probably the situation for the  $\beta$ -phase above 360°C). Therefore the modulation can also be described as a complete ordering of the anion orientations.

The atoms O(1) and O(3) have a large modulation amplitude, and their modulation function  $y^d$  is roughly a square wave. A very schematic projection of the anions in a direction perpendicular to  $\mathbf{p}^*$  and  $\mathbf{b}$  (Fig. 2) illustrates this situation. The complete determination of the modulation amplitudes and wave forms of all atoms in the unit cell will require an elaborate analysis of the satellite intensities. So

far, attempts at such an analysis have shown that the order of magnitude of these intensities does agree with the proposed model.

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